



# Hoberman-sphere-inspired lattice metamaterials with tunable negative thermal expansion

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## ABSTRACT

Materials with engineered thermal expansion coefficients, capable of avoiding failure or irreversible destruction of structures and devices, are important for aerospace, civil, biomedical, optics, and semiconductor applications. In natural materials, thermal expansion usually cannot be adjusted easily and a negative thermal expansion coefficient is still uncommon. Here we propose a novel architected lattice bi-material system, inspired by the Hoberman sphere, showing a wide range of tunable thermal expansion coefficient from negative to positive,  $-1.04 \times 10^{-3} \text{ } ^\circ\text{C}^{-1}$  to  $1.0 \times 10^{-5} \text{ } ^\circ\text{C}^{-1}$ . Numerical simulations and analytical formulations are implemented to quantify the evolution of the thermal expansion coefficients and reveal the underlying mechanisms responsible for this unusual behavior. We show that the thermal expansion coefficient of the proposed metamaterials depends on the thermal expansion coefficient ratio and the axial stiffness ratio of the constituent materials, as well as the bending stiffness and the topological arrangement of the constitutive elements. The finding reported here provides a new routine to design architected metamaterial systems with tunable negative thermal expansion for a wide range of potential applications.

## 1. Introduction

Architected lattice metamaterials with multiple length scales enable the design of new material systems with unprecedented properties, such as negative thermal expansion (NTE) [1,2], negative Poisson's ratio [3–7], ultralightweight and ultrahigh stiffness [8,9], negative stiffness [10], improved thermal resistance [11], and tunable phononic [12] and photonic [13] properties. Especially, their emergence has reactivated researches on NTE materials. Most of natural solid materials have positive thermal expansion, suggesting that they will expand when heated and shrink when cooled due to the elongation or shortening of interatomic bonds. Researchers have found several exceptional natural examples such as zeolites [14], and synthesized composites with NTE, such as  $\text{PbTiO}_3/\text{Cu}$  composites [15] and cement/ $\text{ZrW}_2\text{O}_8$  composites [16], which are usually synthesized by integrating each other in nano- or micro-particles before sintering or solidification. However, they only manifest NTE in a narrow range. Furthermore, these synthesized composites have random microstructures, indicating that the NTE cannot be tailored readily. In order to extensively mitigate the thermal twist, buckle and distortion [17], thermomechanical failure [18], and fatigue [19] from thermal stress mismatch and maintain the geometric stability

[1,2], engineers and material scientists have turned their interests on exploring novel metamaterials with negative thermal expansion. Compared with conventional synthesized composites with random microstructures, lattice metamaterials fabricated by three-dimensional (3D) printing technique and with well-defined topologies can give rise to tunable NTE. The exceptional thermal properties, zero or extremely low thermal expansion, make these metamaterials ideal candidates that can maintain precise dimensional tolerances for many potential engineering applications under large temperature fluctuations in service environments, including aerospace structures, civil engineering structures, optical instruments, biomedical engineering, and semiconductors.

With the rapid progress of additive manufacturing technique, many recent breakthroughs have been achieved in the design of metamaterials from different aspects, such as material compositions, fabrication methods, and modeling approaches. Novel structures of unit cells used in lattice metamaterials have been proposed and can be summarized as: (1) bi-material triangles or the derivatives [2,20–36], (2) chirality [37–41], (3) bilayer buckling origami or trusses [37,42–45], and (4) re-entrant structures [3,28]. As far as compositions of materials are concerned, metal [21,25–31,36–40,46–48], soft polymer [35,38,43,45], composites [2,43] or combination of the aforementioned three [24]

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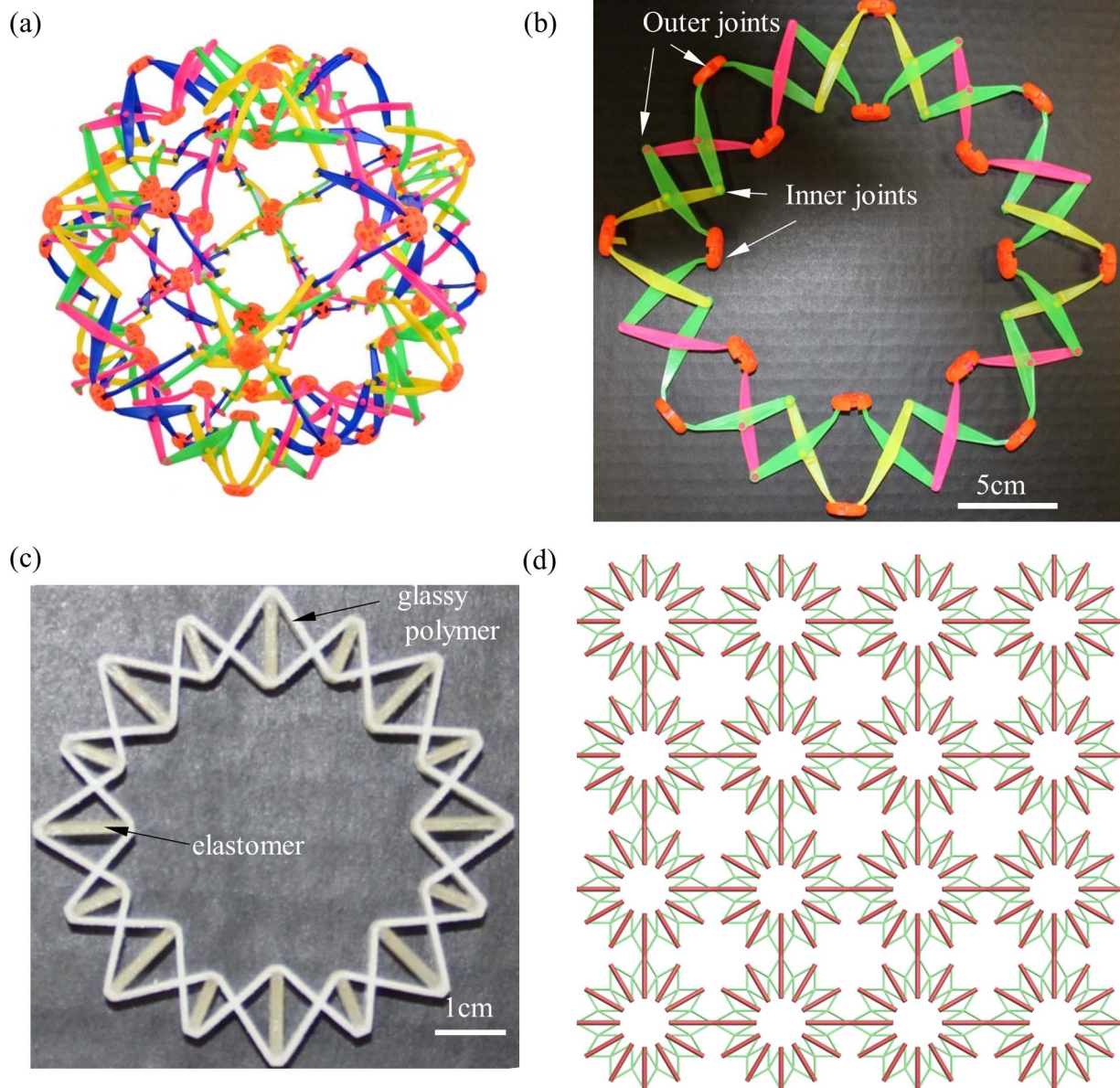


Fig. 1. Design of the lattice metamaterials with tunable thermal expansion: (a) Hoberman expandable sphere, (b) the planar element, (c) the unit cell fabricated by 3D printing technique, and (d) the lattice metamaterial with  $4 \times 4$  unit cells.

have been adopted. In terms of fabrication methods, these metamaterials have been fabricated by welding [21,25,37], micro-stereolithography [2,22], 3D printing [38], laser cut [27], metal lift-off process [46], liquid photoresist and laser lithography [45], and sintering [36]. The thermal expansion coefficient of these metamaterials have also been evaluated through theoretical [23,29,30,32–34,39,40,44,47,49], and numerical [20–22,24,28,31,37,45,48] analyses. Among these artificially designed NTE metamaterials, lattice metamaterials with bi-materials and void space are possible to be adjusted through the volume or area change when the temperature changes. Those researches have enriched tremendously the members of the warehouse of lattice metamaterials with zero or negative thermal expansion. There are other novel existing structures to be referred to pursue NTE like the Hoberman sphere [50]. Moreover, theoretical analysis models and fabrications of those metamaterials with tunable NTE are still evolving, which makes it possible to design more novel NTE structures.

Here, inspired by the Hoberman sphere shown in Fig. 1, we present a class of bi-material lattice metamaterials that can be explored to achieve tunable thermal expansion. The mechanism of the NTE effect depends on its scissors-like microstructure and the circular arrangement of the bi-material triangles. We theoretically and numerically investigate the thermomechanical response of the proposed metamaterials by varying geometric parameters and mechanical properties. A quantitative model considering constituents' axial and bending stiffness is developed to understand the contribution of the NTE mechanism during the thermal variation process. The model also provide the dependence of the NTE on the constituents' mechanical properties and topological features. Finite element simulations are performed to analyze the source of the NTE and are compared with the theoretical model predictions. These models quantify how this negative CTE can be engineered by tailoring the shape, the size, the number of bi-material triangle units, and beams' cross-sectional geometry, providing a viable avenue to analyze the effective CTE of other NTE structures.

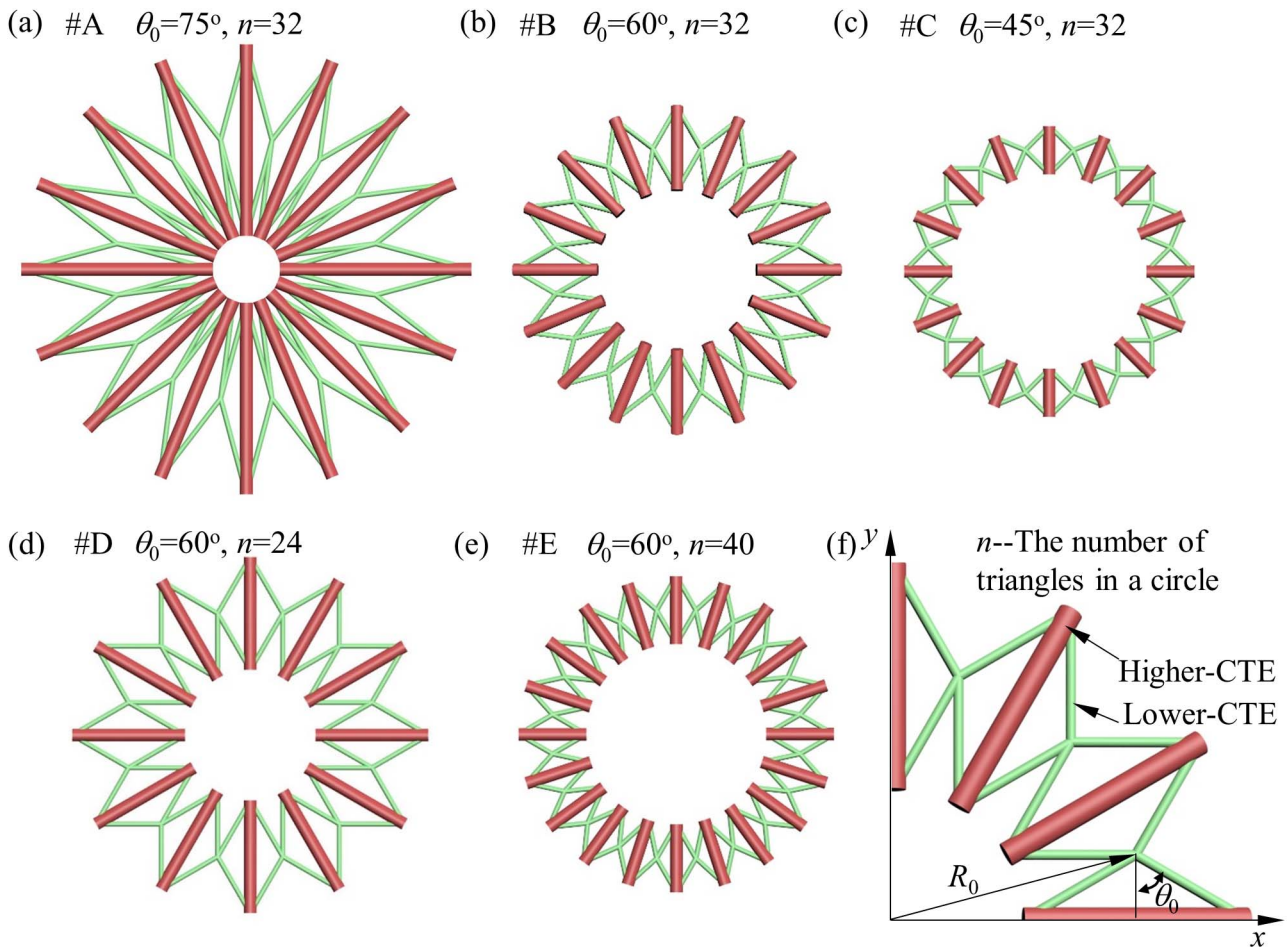


Fig. 2. Representative unit cells of the proposed lattice metamaterials. Those unit cells including (a)–(e) are characterized by the radius of the circle (distance between the center and green cross point)  $R_0$  with  $R_0 = 10$  mm, the half vertex angle between blue beams in the triangle  $\theta_0$ , and the number of triangles in a circle,  $n$ . (f) One quarter unit cell with definitions including  $R_0$ ,  $\theta_0$ , and  $n$ , as well as Higher-CTE (the material of higher coefficient of thermal expansion), Lower-CTE (the material of lower coefficient of thermal expansion) constituents. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

## 2. Models and methods

### 2.1. Characterization of lattice metamaterials

The design concept is inspired by the Hoberman’s expandable sphere (see Fig. 1(a)), which is an isokinetic structure that resembles a geodesic dome. It is capable of folding down to a fraction of its normal size by the scissor-like action of the joints. When it contracts, the distance of outer and inner joints elongates. Referring to its geometric features and deformation mechanisms, we propose a class of unit cells with NTE, which inherit scissor-like geodesic circle (see Fig. 1(b)). We append a linking beam in the outer and inner joints and change hinges into fixed joints, as shown in Fig. 1(c) where the designed unit cell is fabricated using 3D printing technique. The unit cells are characterized by the radius of circular arrangement,  $R_0$ , the number of triangles,  $n$ , and half vertex angle of the triangle,  $\theta_0$ , as shown in Fig. 2. As  $R_0$ ,  $n$ , and  $\theta_0$  vary, we can construct a series of unit cells, #A to #E.

The unit cell of the proposed lattice metamaterials is made up of  $n$  totally identical bi-material isosceles triangles (see Figs. 2 and 3), one of which is composed of a base member (red beam) with a higher coefficient of thermal expansion (higher-CTE),  $\alpha_h$ , and two legs (blue beams) with a lower coefficient of thermal expansion (lower-CTE),  $\alpha_l$  (Fig. 3). Those triangles are evenly distributed in a polar pattern around a center point, and some of them share a base, while others have a common vertex. Blue and red beams expand at different rates when heated: the blue beams are induced to fold toward the red beams, compressing the voids and ultimately occupying less space, vice versa,

as shown in Fig. 3(a)–(b). Because of the circular pattern, the circumferential compression squeezes further the central voids, which magnifies the strain rate of the unit cell, vice versa.

### 2.2. Analytical model

Since the idealized circle is centrally symmetrical, it can be divided into  $n$  segments of equal arcs, one of which is approximately equal to the height of the triangle,  $h_0$ . It is the fundamental module characterized by minimum contracted or expanded element illustrated in Fig. 3(c). When the temperature changes from  $T_0$  to  $T_0 + \Delta T$ , the shrinkage or elongation of the circular arrangement, or triangles ring chain from  $nh_0$  to  $nh$  leads to its own circumference longer or shorter. To retain the entire circular shape, radius reduction or expansion appears from  $R_0$  to  $R$ , which indicates the negative thermal expansion in the proposed metamaterials. To avoid local buckling in compressive beams, we limit  $\theta_0$  in the interval  $[30^\circ, 75^\circ]$ . We assume that (1) all the constituents are elastic, recoverable, and small deformable materials, and (2) during the thermal deformation, the angle  $\varphi$  keeps approximately constant as shown in Fig. 3(c).

In the representative isosceles triangle in Fig. 3(c), its height, the length of its base or higher-CTE beam, and the length of its leg or lower-CTE beam are  $h_0 = R_0 \sin \varphi$ ,  $l_0 = 2R_0 \sin \varphi \tan \theta_0$ , and  $m_0 = \frac{R_0 \sin \varphi}{\cos \theta_0}$ , respectively. The initial distance between the center and the outer endpoint  $S_0$  is expressed as

$$s_0 = R_0 (\cos \varphi + \sin \varphi \tan \theta_0), \tag{1}$$

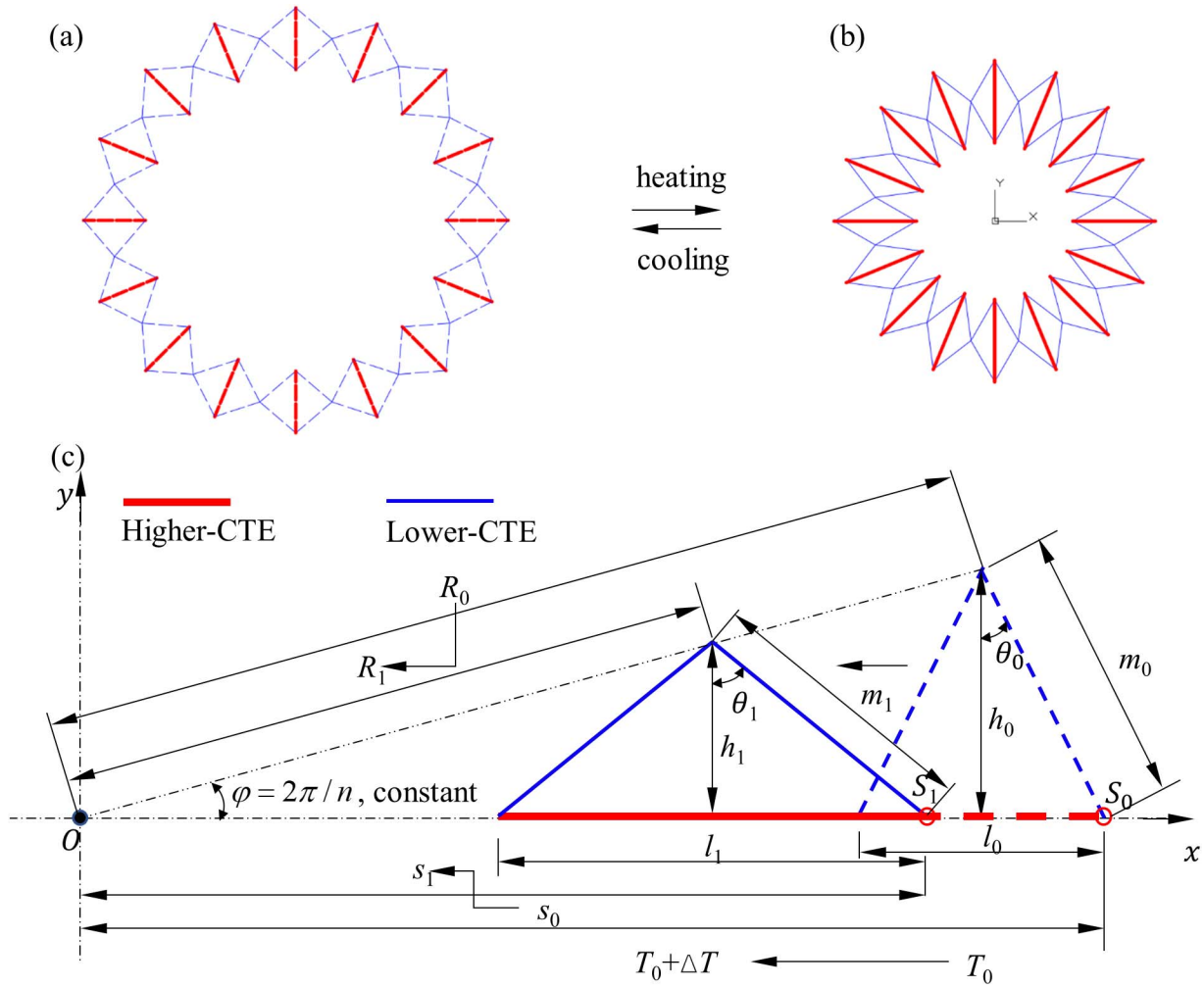


Fig. 3. Theoretical model of thermal deformation of the unit cell. (a) The unit cell before heated, (b) the unit cell after heated, and (c) the representative element of the unit cell, or bi-material isosceles triangle, where  $T_0$  and  $T_0 + \Delta T$  denote the initial temperature, and the temperature after heated, respectively.  $R_0$  is the initial radius of the circle,  $\theta_0$  the initial angle between the height and the leg of the triangle,  $m_0$  the initial length of the leg of the triangle,  $l_0$  initial length of base line of the triangle,  $s_0$  the initial distance between the origin  $O$  and the outer endpoint  $S_0$ ,  $h_0$  the initial height of the triangle. When the temperature changes,  $R_0$ ,  $\theta_0$ ,  $m_0$ ,  $l_0$ ,  $s_0$ , and  $h_0$  change into  $R$ ,  $\theta$ ,  $m$ ,  $l$ ,  $s$ , and  $h$ , respectively.

where the central angle  $\varphi = 2\pi/n$ , keeps approximately constant. The distance between the center and the outer endpoint  $S$  changes into

$$s = s_0 + u_1, \tag{2}$$

where  $u_1$  is the horizontal displacement of the outer point  $S$ . And the strain of the entire unit cell is described isotopically as

$$\varepsilon = \frac{s - s_0}{s_0}. \tag{3}$$

Substituting Eqs. (1)–(3) to  $\alpha = \frac{\varepsilon}{\Delta T}$ , we can obtain effective CTE as

$$\alpha = \frac{1}{\Delta T} \frac{u_1}{R_0(\cos\varphi + \sin\varphi \tan\theta_0)}, \tag{4}$$

where  $u_1$  is the key factor which solves  $\alpha$ . Due to highly central symmetry, the solution of  $u_1$  relies on the representative isosceles triangle, which consists of three clamped beams with symmetric boundary conditions along the axes  $OX$  and  $OG$  as shown in Fig. 4(c). It can be solved by the displacement method in structural mechanics using beam element in Fig. 4(a). Nodal forces and moments are denoted as  $\{U_i \ V_i \ M_i\}^T$ , and  $\{U_j \ V_j \ M_j\}^T$ , and nodal displacements and rotations as  $\{u_i \ v_i \ \varphi_i\}^T$ , and  $\{u_j \ v_j \ \varphi_j\}^T$  at Node  $i$  and  $j$  in a beam element on the local coordinate system. The element local stiffness matrix is conducted as

$$[k]^e = \begin{bmatrix} \frac{AE}{l} & 0 & 0 & -\frac{AE}{l} & 0 & 0 \\ 0 & \frac{12EI}{l^3} & \frac{6EI}{l^2} & 0 & \frac{-12EI}{l^3} & \frac{6EI}{l^2} \\ 0 & \frac{6EI}{l^2} & \frac{4EI}{l} & 0 & \frac{-6EI}{l^2} & \frac{2EI}{l} \\ -\frac{AE}{l} & 0 & 0 & \frac{AE}{l} & 0 & 0 \\ 0 & \frac{-12EI}{l^3} & \frac{-6EI}{l^2} & 0 & \frac{12EI}{l^3} & \frac{-6EI}{l^2} \\ 0 & \frac{6EI}{l^2} & \frac{2EI}{l} & 0 & \frac{-6EI}{l^2} & \frac{4EI}{l} \end{bmatrix}, \tag{5}$$

where  $A$  is the sectional area,  $E$  is the Young's Modulus,  $I$  is the inertia moment, and  $l$  is the length of the beam element in the local coordinate system in Fig. 4(a).

The element nodal forces and displacements meet

$$\{F\} = [k]^e \{\delta\}, \tag{6}$$

where the element nodal force is  $\{F\} = \{U_i \ V_i \ M_i \ U_j \ V_j \ M_j\}^T$ , and the element nodal displacement is  $\{\delta\} = \{u_i \ v_i \ \varphi_i \ u_j \ v_j \ \varphi_j\}$  as shown in Fig. 4(a). When the global coordinate system is transformed from the local coordinate system as shown in Fig. 4(b), the element global stiffness matrix is expressed as

$$[K]^e = [h]^T [k]^e [h], \tag{7}$$

where  $[h]$  is the coordinate transformation matrix as

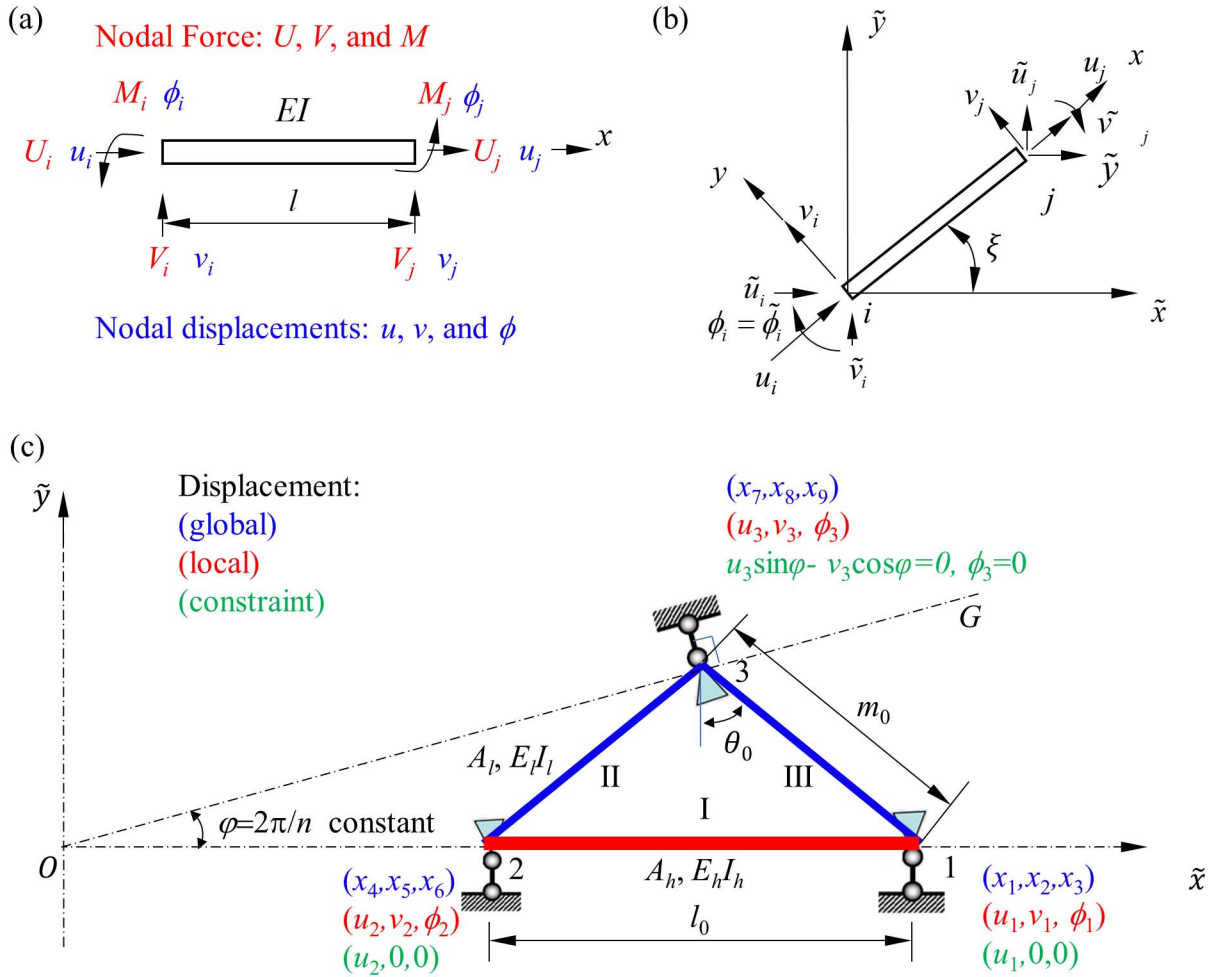


Fig. 4. Beam element and representative isosceles triangle frame with boundary conditions. (a) Beam element with nodal displacements, rotations, forces, and moments. (b) Conversion from the local system to the global system. (c) Elements, nodes, and constraint conditions. The global elements are I, II and III. The global nodes are designated as 1, 2 and 3. The global nodal displacements are  $x_1, x_2, x_3, x_4, x_5, x_6, x_7, x_8$  and  $x_9$ .  $A_b, E_b$  and  $I_b$  are the sectional area, Young's Modulus and the inertia moment of the lower-CTE beam, respectively.  $A_h, E_h$  and  $I_h$  are the sectional area, Young's Modulus and the inertia moment of the higher-CTE beam, respectively.

$$[h] = \begin{bmatrix} \cos \xi & \sin \xi & 0 & 0 & 0 & 0 \\ -\sin \xi & \cos \xi & 0 & 0 & 0 & 0 \\ 0 & 0 & 1 & 0 & 0 & 0 \\ 0 & 0 & 0 & \cos \xi & \sin \xi & 0 \\ 0 & 0 & 0 & -\sin \xi & \cos \xi & 0 \\ 0 & 0 & 0 & 0 & 0 & 1 \end{bmatrix}, \quad (8)$$

and  $\xi$  is the angle between local and global  $x$ -axis along the clockwise direction. The three elements' stiffness matrices are listed as Appendix A. The global stiffness matrix  $[K]$  is assembled as Appendix (A4).

As shown in Fig. 4(c), the triangular frame's boundary conditions are described as

$$\begin{cases} x_2 = 0; x_3 = 0; \\ x_5 = 0; x_6 = 0; \\ x_7 \sin \varphi - x_8 \cos \varphi = 0; x_9 = 0; \end{cases}, \quad (9)$$

where  $\{X\} = \{x_1 \ x_2 \ x_3 \ x_4 \ x_5 \ x_6 \ x_7 \ x_8 \ x_9\}^T$  defines global nodal displacements. Element local thermal load,  $\{P_T^e\} = \{EA\alpha\Delta T \ 0 \ 0 \ -EA\alpha\Delta T \ 0 \ 0\}^T$ , is transformed to the global coordinate system by  $\{P_T\} = [h]^T \{P_T^e\}$ . Assembling global stiffness matrix and calculating thermal loads, we produce a  $9 \times 9$  symmetric linear equations as  $[K]\{u\} = \{P\}$ . Dealing with the boundary conditions, we obtain a solution with respect to  $\{u_1 u_2 u_3\}^T$ , which is the solution of  $[K']\{u\} = \{P'\}$  or

$$\begin{bmatrix} K'_{11} & K'_{12} & K'_{13} \\ K'_{21} & K'_{22} & K'_{23} \\ K'_{31} & K'_{32} & K'_{33} \end{bmatrix} \begin{Bmatrix} u_1 \\ u_2 \\ u_3 \end{Bmatrix} = \begin{Bmatrix} P'_1 \\ P'_2 \\ P'_3 \end{Bmatrix}. \quad (10)$$

More details are listed in Appendix A.

After solving the linear equations, and substituting them to Eq. (4), we can calculate the effective CTE. So, Eqs. (4) and (9) are used to describe the effective CTE of proposed metamaterials. The LU Decomposition Method is adopted to solve Eq. (9) using FORTRAN. The effective CTE,  $\alpha$ , is associated with  $E_h, A_h, \alpha_h, E_b, A_b, I_b, \alpha_b, \theta_0$  and  $\varphi$ . Furthermore, it is relevant to the CTE ratio of the Higher-CTE beam and the lower-CTE beam,  $\frac{\alpha_h}{\alpha_l}$ , and the axial stiffness ratio,  $\frac{A_h E_h}{A_l E_l}$  of Higher-CTE and Lower-CTE beams, as well as bending stiffness,  $\frac{I_l}{A_l}$  of the lower-CTE beam, and irrelevant to increment temperature,  $\Delta T$ , within the linear elastic range.

### 2.3. Finite element simulations

To verify the theoretical models and better understand the behavior of the proposed metamaterials, we investigate their thermal responses through numerical simulations with the finite element Abaqus/Standard package. In all simulations, we consider a quarter of a unit cell with suitable periodic boundary conditions. 2D solid models are constructed using 4-node bilinear plane stress element CPS4R. No out-of-plane deformation is allowed in the current study. Thermal load is considered for the temperature change from 20 °C to 70 °C.

**Table 1**  
Mechanical properties of adopted bi-material combinations.

Higher/Lower-CTE	PEGDA/Cu-PEGDA	Teflon/Acrylic	Al/Ti	Al/Invar	A/B
$E_h, E_l$ (GPa)	0.0235,	0.475, 3.2	70, 116	70, 144	10, 10
$\alpha_h, \alpha_l$ ( $10^{-6} \text{ } ^\circ\text{C}^{-1}$ )	0.1124	123, 67	23.1, 8.6	23.1, 1.1	200, 20
	155.6, 39.8				
$E_h/E_l$	0.21	0.15	0.603	0.486	1
$\alpha_h/\alpha_l$	3.91	1.84	2.686	21	10

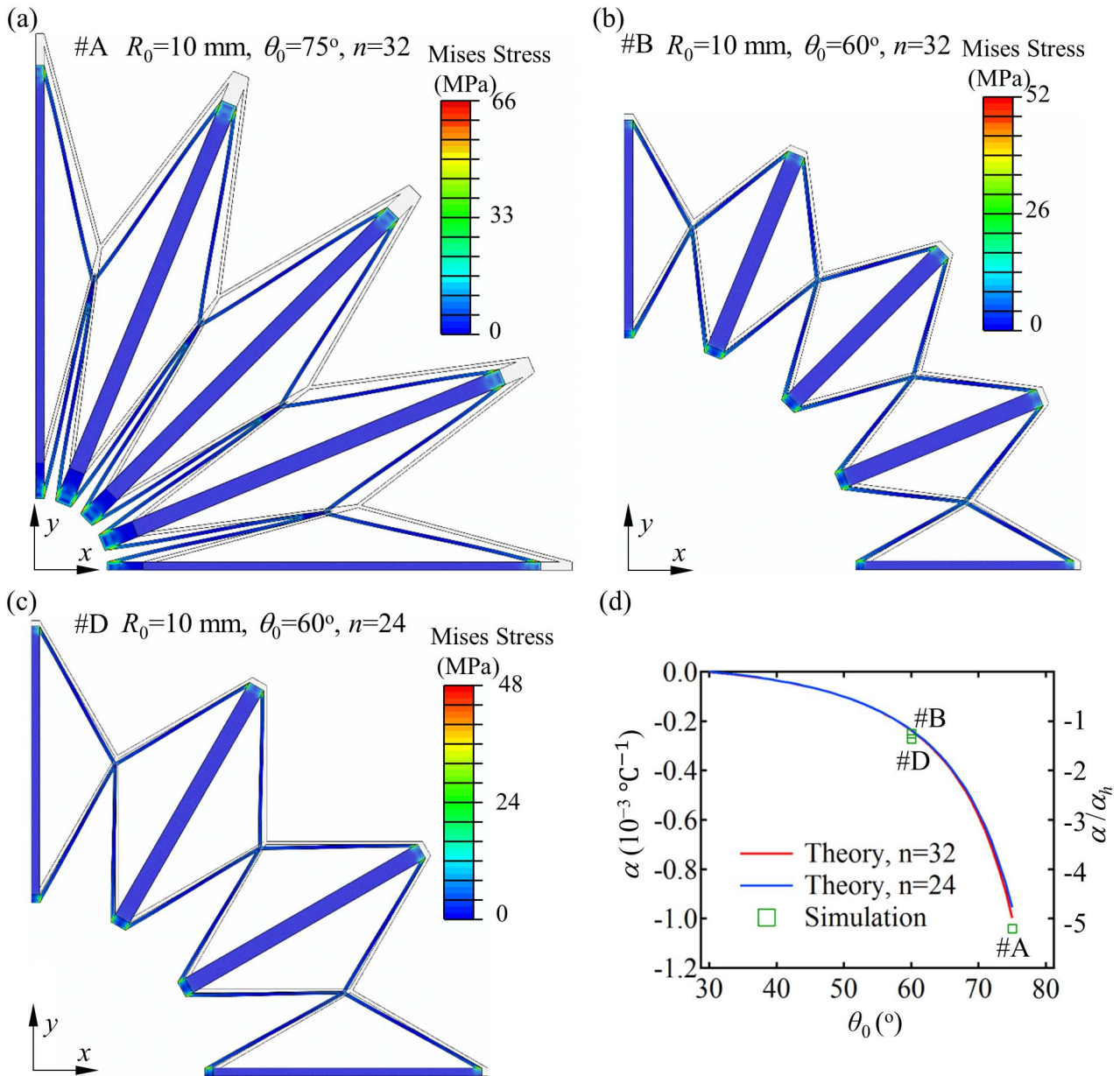
Parametric studies in geometry and mechanical properties have been conducted using the unit cell-based finite element models and theoretical models. We build models of the unit cells with  $R_0 = 10 \text{ mm}$ ,  $24 \leq n \leq 40$ , and  $30^\circ \leq \theta_0 \leq 75^\circ$ , and compare numerical and theoretical results based on possible constituents mentioned in the literature

[2,27,28,51]. Table 1 lists several material combinations and customized materials (A/B) with  $E_h=10 \text{ GPa}$ ,  $\alpha_h = 2 \times 10^{-4} \text{ } ^\circ\text{C}^{-1}$ ,  $E_l = 10 \text{ GPa}$ ,  $\alpha_l = 2 \times 10^{-5} \text{ } ^\circ\text{C}^{-1}$ , and  $\nu_h = \nu_l = 0.3$ . We also vary the ratio of two constituents' CTE  $\alpha_h/\alpha_l$  from 10.0 to 1.0, the ratio of two constituents' sectional area,  $A_h/A_l$  from 10.0 to 0.1, and the ratio of two constituents' Young's Modulus  $E_h/E_l$  from 0.001 to 1000 in our parametric studies.

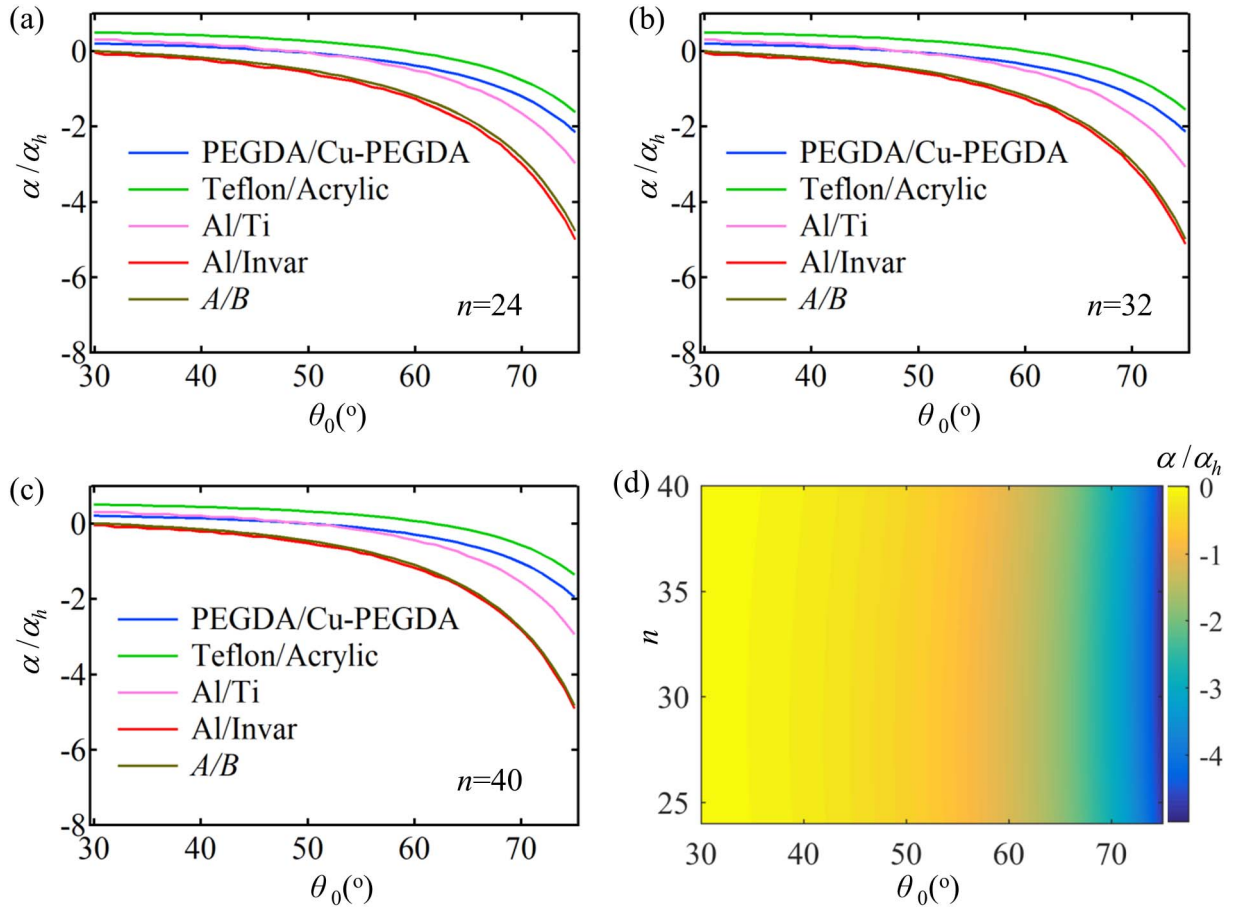
**3. Results and discussion**

*3.1. Existence of negative thermal expansion*

Fig. 5 shows a good agreement between theoretical and numerical results, although the elastic 2D solid element deforms essentially more than the beam element. Fig. 5(a)–(c) displays that NTE phenomenon clearly occur in the unit cells #A, #B, and #D. As the temperature increases, all higher-CTE beams moving inward actuate the shrinkage



**Fig. 5.** Mises stress contours as well as numerical and theoretical comparison of a quarter of cooled metamaterial unit cells including #A in (a), #B in (b), #D in (c), as  $\Delta T = 50 \text{ } ^\circ\text{C}$ ,  $A_h/A_l = 5$ ,  $\alpha_h/\alpha_l = 10$ ,  $E_h/E_l = 1$  and  $\alpha_h = 2 \times 10^{-4} \text{ } ^\circ\text{C}^{-1}$ . (d) The effective CTE and normalized effective CTE of the unit cells with  $30^\circ \leq \theta_0 \leq 75^\circ$  as well as  $n = 24$  and  $n = 32$  using theoretical models and numerical simulations. The color scale bars of Mises stress are linear.



**Fig. 6.** The normalized effective CTE of the lattice metamaterials with  $\theta_0 \in [30^\circ, 75^\circ]$ ,  $n \in [24, 40]$ ,  $\Delta T = 50^\circ\text{C}$ , using different bi-material combinations. (a) Normalized effective CTE versus  $\theta_0$  with  $n = 24$ ,  $30^\circ \leq \theta_0 \leq 75^\circ$ . (b) Normalized effective CTE versus  $\theta_0$  with  $n = 32$ ,  $30^\circ \leq \theta_0 \leq 75^\circ$ . (c) Normalized effective CTE versus  $\theta_0$  with  $n = 40$ ,  $30^\circ \leq \theta_0 \leq 75^\circ$ . (d) Normalized effective CTE using customized materials (A/B) with  $E_h = 10\text{ GPa}$ ,  $\alpha_h = 2 \times 10^{-4}\text{ }^\circ\text{C}^{-1}$ , as well as  $E_l = 10\text{ GPa}$ ,  $\alpha_l = 2 \times 10^{-5}\text{ }^\circ\text{C}^{-1}$ .

of the entire unit cell. As shown in Fig. 5(d), both numerical analysis and theoretical models show that the value of NTE rises with the increase of the angle  $\theta_0$ . As expected, the thermal expansion is almost independent of  $n$ . The maximum NTE happens with normalized effective  $\alpha/\alpha_h = -5.2$  and  $\alpha = -1.04 \times 10^{-3}\text{ }^\circ\text{C}^{-1}$  when  $\theta_0$  of the unit cell is  $75^\circ$ .

The theoretical and numerical investigations verify that tunable or negative thermal expansion does occur among those proposed metamaterials in a wide range. The elongation or contraction of the higher-CTE beams induces triangles flattened or sharpened, which causes the triangle circumference shorter or longer after heated or cooled. The NTE effect of this scissors-like and ring-like configuration is clearly demonstrated in this study.

### 3.2. Effect of geometric parameters

The NTE in the proposed metamaterials provides us flexible space on selecting suitable constituents. We chose alternative metallic and polymeric materials mentioned above to investigate the correlation between NTE and geometric parameters.

Fig. 6 shows the normalized effective CTE of the lattice metamaterials considering different bi-material combinations. Note that when  $\theta_0$  is small, the metamaterials with three material combinations (Teflon/Acrylic, Al/Ti, and PEGDA/Cu-PEGDA) have positive CTEs. When  $\theta_0$  is greater than  $50^\circ$ , the CTE becomes negative. However, the metamaterials with A/B and Al/Invar combinations exhibit negative CTEs with  $\theta_0$  from  $30^\circ$  to  $75^\circ$ , and decreases as  $\theta_0$  increases. Fig. 6(d) provides the contour of the normalized effective CTE of the lattice metamaterials with  $30^\circ \leq \theta_0 \leq 75^\circ$ ,  $24 \leq n \leq 40$  using customized materials (A/B),

indicating the impact of  $\theta_0$  and  $n$  on the normalized effective CTE. Clearly, the angle  $\theta_0$  plays a more significant role in the negative CTE, indicating tuning of CTE with well-designed angle  $\theta_0$ . The number  $n$  has less effect on the CTEs, which is confirmed in Fig. 7. In general, the effective CTE is more sensitive to  $\theta_0$  than  $n$ , while it is irrelevant to  $R_0$ . It's also observed that when  $\theta_0$  is smaller than  $60^\circ$ , the normalized CTE of all metamaterials is between  $-2$  and  $0.2$ . However, when  $\theta_0$  is larger such as  $75^\circ$  as shown in Fig. 7(c), the normalized CTE can be small from  $-5.2$  to  $-1.04 \times 10^{-3}\text{ }^\circ\text{C}^{-1}$ . The metamaterial with Teflon/Acrylic combination always has larger CTEs, but the metamaterial with Al/Invar combination has the smallest CTEs due to the higher ratio of CTEs.

### 3.3. Effect of mechanical properties

Considering that the normalized effective CTE of #A is the smallest (largest in absolute value) among all unit cells listed in Fig. 2, we take metamaterial #A as a representative example to analyze the effect of the mechanical properties on the CTE. Fig. 8 depicts the evolution of the effective CTE of metamaterial #A as a function of CTE ratio  $\alpha_h/\alpha_l$  and the ratio of Young's modulus,  $E_h/E_l$ , where  $\Delta T = 50^\circ\text{C}$ . Clearly the normalized effective CTE strongly depends on both CTE ratio and Young's modulus ratio. When Young's modulus ratio is from  $E_h/E_l = 0.001$  to  $E_h/E_l = 0.01$ , the normalized effective CTE is close to 0 for all  $\alpha_h/\alpha_l$  from 1 to 10. When  $E_h/E_l$  is larger than 0.01, the normalized effective CTE is positive when  $\alpha_h/\alpha_l$  is smaller than  $\sim 1.5$ ; however, the normalized effective CTE turns negative when  $\alpha_h/\alpha_l$  is larger than  $\sim 1.5$ . Higher  $\alpha_h/\alpha_l$  indicates lower effective CTE. A minimum normalized effective CTE  $\alpha/\alpha_h$  emerges at  $-5.2$  and  $\alpha = -1.04 \times 10^{-3}\text{ }^\circ\text{C}^{-1}$  as

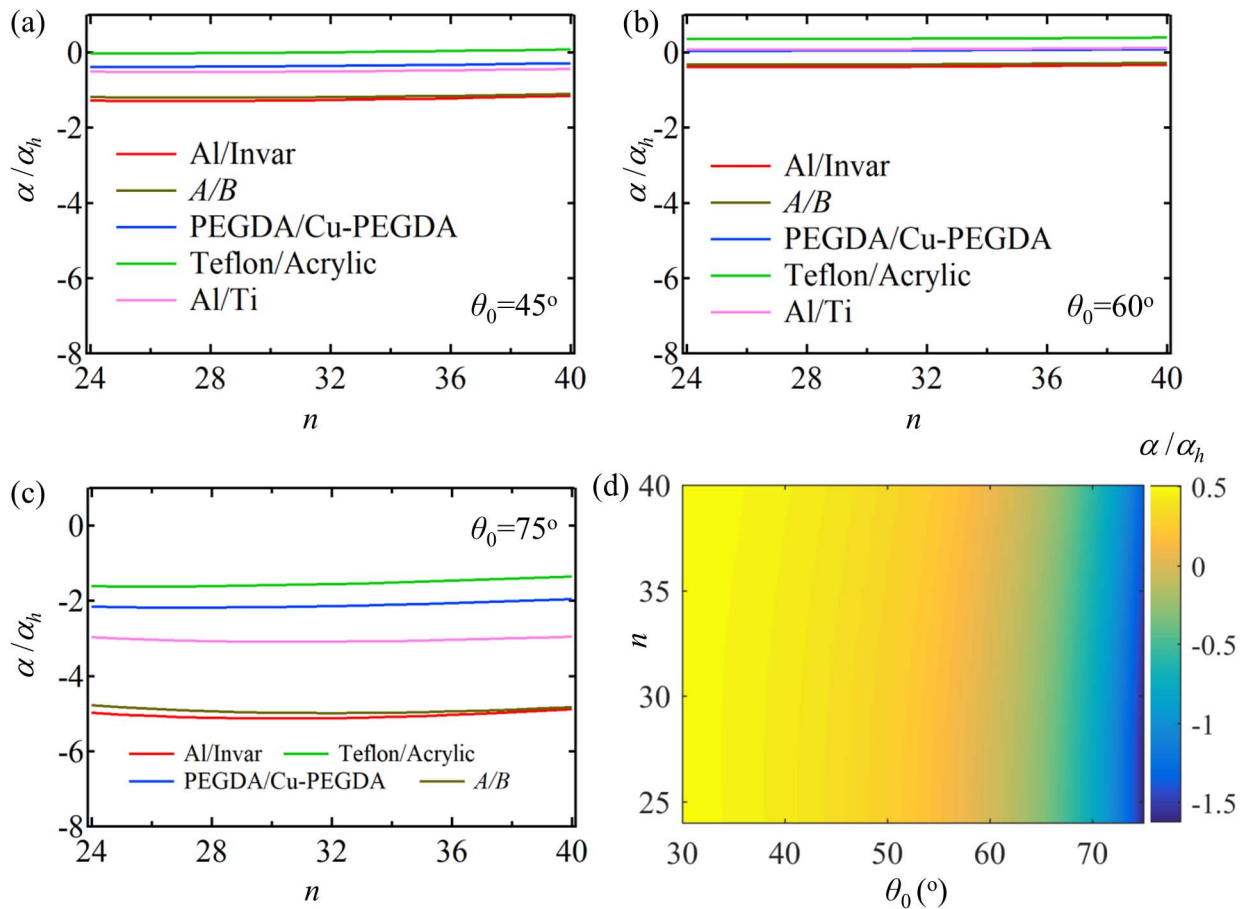


Fig. 7. The normalized effective CTE of the lattice metamaterials with  $\theta_0 \in [30^\circ, 75^\circ]$ ,  $n \in [24, 40]$ ,  $\Delta T = 50^\circ\text{C}$ , using different bi-material combination. (a) Normalized effective CTE versus  $n$  with  $\theta_0 = 45^\circ$ ,  $24 \leq n \leq 40$ . (b) Normalized effective CTE versus  $n$  with  $\theta_0 = 60^\circ$ ,  $24 \leq n \leq 40$ . (c) Normalized effective CTE versus  $n$  with  $\theta_0 = 75^\circ$ ,  $24 \leq n \leq 40$ . (d) Normalized effective CTE as  $30^\circ \leq \theta_0 \leq 75^\circ$ ,  $24 \leq n \leq 40$  using Teflon/Acrylic.

shown in Fig. 8(b). In Fig. 8(c)–(d), for metamaterial #A with customized materials, we present the evolution of the normalized effective CTE as the function of the CTE ratio  $\alpha_h/\alpha_l$  and the sectional area ratio  $A_h/A_l$ , while  $E_h/E_l = 1$  and  $\Delta T = 50^\circ\text{C}$ . Similarly, for the sectional area ratio from 0.1 to 10, the normalized effective CTE is positive when  $\alpha_h/\alpha_l$  is smaller than  $\sim 1.5$ ; however, the normalized effective CTE turns negative when  $\alpha_h/\alpha_l$  is larger than  $\sim 1.5$ . Dramatic changes are found for  $\alpha_h/\alpha_l$  increasing from 1 to 4.

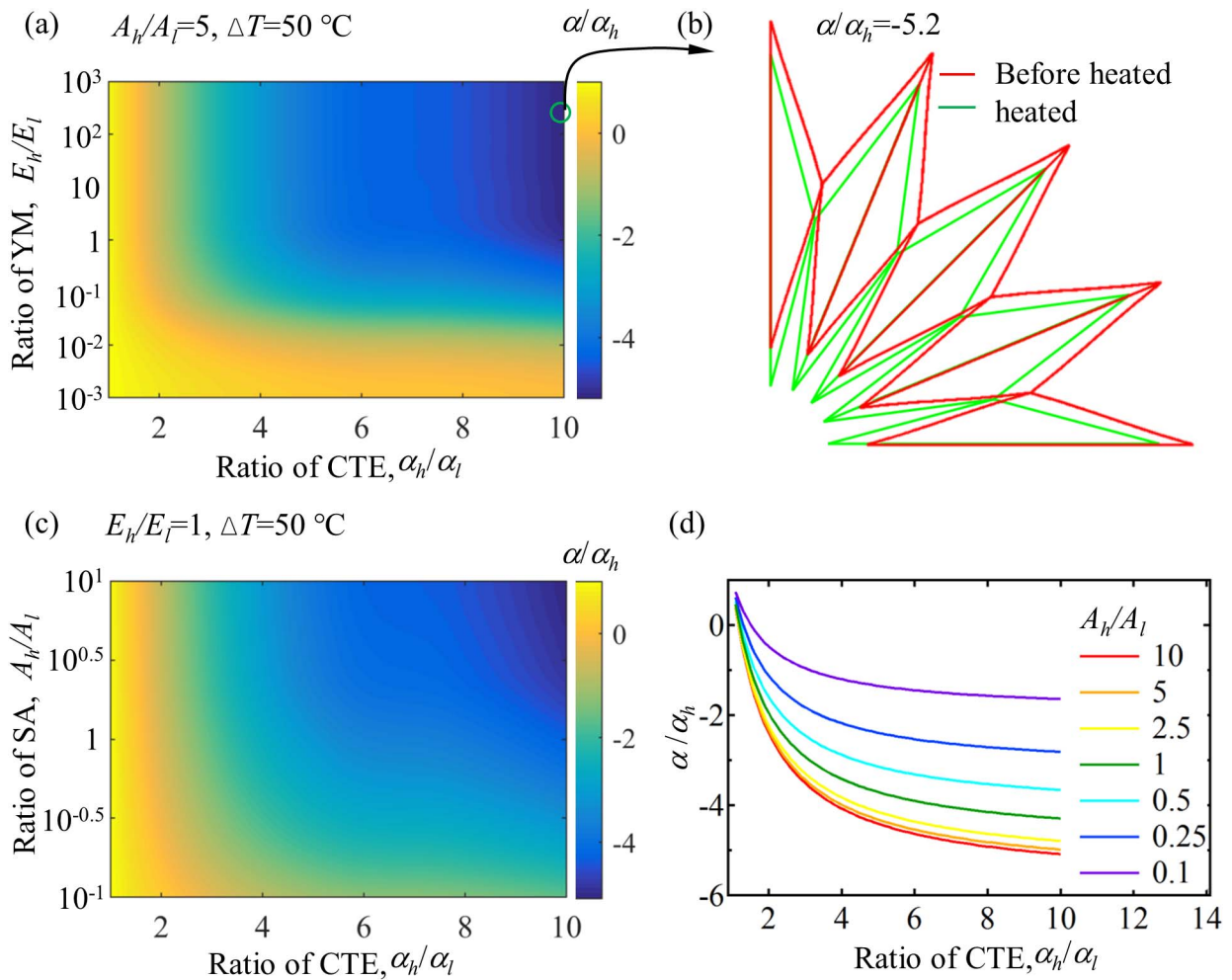
To verify the results experimentally, a heating process (from  $22.5^\circ\text{C}$  to  $68.0^\circ\text{C}$ ) is performed on the unit cell (see Fig. 1(c)) fabricated by a multiple material 3D printer (Objet260 Connex, Stratasys Ltd), where  $A_h/A_l = 2$ ,  $\alpha_h/\alpha_l = 1.5$  and  $E_h/E_l = 0.0034$ . The measured CTE of this sample is  $2.9 \times 10^{-4} \text{ }^\circ\text{C}^{-1}$ , which is consistent with the effective CTE as  $2.2 \times 10^{-4} \text{ }^\circ\text{C}^{-1}$  predicted by our model, indicating a validation of our analytical model and simulation method. Note that this sample shows positive CTE, which is limited by the materials from our 3D printing method. However, by tuning the geometrical parameters and constituent materials' properties, the desired effective CTE of the bi-material lattice metamaterials can be achieved. In general, the primary influencing factor is the CTE ratio of higher-CTE and lower-CTE constituents; the secondary one is the sectional area ratio; and the third one is the ratio of Young's modulus. This finding suggests that the CTE can be tailored to be positive, zero, and even negative by controlling these effects, thereby offering opportunities to design lattice metamaterials with tunable CTE.

Furthermore, the stiffness of the NTE structures is crucial for potential applications. To study the mechanical properties especially the stiffness of the proposed metamaterials, uniaxial tension is carried out using finite element analysis. The details of the simulation can be found

in Appendix B. The calculated Young's Modulus is shown in Table 2. Compared with their constituent materials, the proposed metamaterials exhibit relatively low stiffness because of their bending-dominated deformations of the lattice structures. However, they still have considerable load-carrying capability in addition to the unusual NTE property. Further work is needed to investigate the mechanical response of the proposed lattice metamaterials and seek the trade-off between the NTE and the stiffness in applications.

#### 4. Conclusion

In summary, we have demonstrated an approach to create bi-material lattice metamaterials with tunable negative or extremely low thermal expansion coefficient from  $-1.04 \times 10^{-3} \text{ }^\circ\text{C}^{-1}$  to  $1.0 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$ , inspired by the Hoberman sphere. Through a combination of analytical analysis and numerical simulations, we have showed that the thermal expansion coefficient of the bi-material lattice metamaterials can be tuned by selecting the constituent materials with different thermal expansion coefficients, stiffness (tension/compression and bending), and the geometric parameters of the lattice metamaterials. In particular, the quantitative modeling of thermo-mechanical behavior presented here can precisely predict the thermal expansion coefficient of the bi-material lattice metamaterials. It can be envisioned that the proposed design strategy can be extended to 1D and 3D metamaterials, providing insights into the development of a new class of architected metamaterials with potential applications including aerospace industry, civil engineering, biomedical sensors, optics, and semiconductors.



**Fig. 8.** The effect of mechanical properties on the CTE of metamaterial #A ( $R = 10 \text{ mm}$ ,  $\theta_0 = 75^\circ$ ,  $n = 32$ ). (a) Effective CTE of #A between  $\alpha/\alpha_h$  versus the ratio of CTE  $\alpha_h/\alpha_l$ , the ratio of Young's Modulus,  $E_h/E_l$ , and (b) the corresponding thermally deformed shape in theoretical model. (c) Normalized effective CTE of #A between  $\alpha/\alpha_h$  versus the ratio of CTE  $\alpha_h/\alpha_l$ , ratio of stiffness  $A_h/A_l$ . (d) Normalized effective CTE  $\alpha/\alpha_h$ , and the ratio of CTE  $\alpha_h/\alpha_l$ , of #A with the ratio of sectional area  $A_h/A_l=10, 5, 2.5, 1, 0.5, 0.25$  and  $0.1$ .

**Table 2**

The equivalent Young's modulus of the bi-material metamaterials ( $\bar{V}$  is the volume fraction of the unit cell,  $\bar{E}$  is the equivalent Young's modulus of the metamaterials, and  $E_0$  is the Young's Modulus of the Lower-CTE material, MPa).

bi-material	PEGDA/Cu-PEGDA			Teflon/Acrylic			Al/Ti		Al/Invar		A/B	
Unit cell	$\bar{V}$	$\bar{E}$	$E_0\bar{V}$	$\bar{E}$	$E_0\bar{V}$	$\bar{E}$	$E_0\bar{V}$	$\bar{E}$	$E_0\bar{V}$	$\bar{E}$	$E_0\bar{V}$	
#A	0.2045	$3.5 \times 10^{-3}$	22	0.084	654.3	3.61	$23.7 \times 10^3$	3.76	$29.43 \times 10^3$	0.32	2044	
#B	0.1349	$4.6 \times 10^{-3}$	15.4	0.13	438.2	4.76	$16 \times 10^3$	5.9	$19.7 \times 10^3$	0.41	1350	
#D	0.1248	$4.8 \times 10^{-3}$	14	0.14	399.4	5.0	$14 \times 10^3$	6.2	$18 \times 10^3$	0.43	1248	

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**Appendix A**

The stiffness matrix of Element I is:

$$[K]_{II}^e = \begin{bmatrix} \frac{A_h E_h}{l_0} & 0 & 0 & -\frac{A_h E_h}{l_0} & 0 & 0 \\ 0 & \frac{12E_h I_h}{l_0^3} & \frac{-6E_h I_h}{m_0^2} & 0 & \frac{-12E_h I_h}{l_0^3} & \frac{-6E_h I_h}{l_0^2} \\ 0 & \frac{-6E_h I_h}{l_0^2} & \frac{4E_h I_h}{l_0} & 0 & \frac{6E_h I_h}{l_0^2} & \frac{2E_h I_h}{l_0} \\ -\frac{A_h E_h}{l_0} & 0 & 0 & \frac{A_h E_h}{l_0} & 0 & 0 \\ 0 & \frac{-12E_h I_h}{l_0^3} & \frac{6E_h I_h}{m_0^2} & 0 & \frac{12E_h I_h}{l_0^3} & \frac{6E_h I_h}{l_0^2} \\ 0 & \frac{-6E_h I_h}{l_0^2} & \frac{2E_h I_h}{l_0} & 0 & \frac{6E_h I_h}{l_0^2} & \frac{4E_h I_h}{l_0} \end{bmatrix} \quad (A1)$$

The stiffness matrix of Element II is:

$$[k]_{II}^e = \begin{bmatrix} \frac{A_l E_l}{m_0} \sin^2 \theta_0 + \frac{12E_l I_l}{m_0^3} \cos^2 \theta_0 & \left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & -\frac{6E_l I_l}{m_0^2} \cos \theta_0 - \frac{A_l E_l}{m_0} \sin^2 \theta_0 - \frac{12E_l I_l}{m_0^3} \cos^2 \theta_0 & -\left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & -\frac{6E_l I_l}{m_0^2} \cos \theta_0 \\ \left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & \frac{A_l E_l}{m_0} \cos^2 \theta_0 + \frac{12E_l I_l}{m_0^3} \sin^2 \theta_0 & \frac{6E_l I_l}{m_0^2} \sin \theta_0 & -\left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & -\frac{A_l E_l}{m_0} \cos^2 \theta_0 - \frac{12E_l I_l}{m_0^3} \sin^2 \theta_0 & \frac{6E_l I_l}{m_0^2} \sin \theta_0 \\ -\frac{6E_l I_l}{m_0^2} \cos \theta_0 & \frac{6E_l I_l}{m_0^2} \sin \theta_0 & \frac{4E_l I_l}{m_0} & \frac{6E_l I_l}{m_0^2} \cos \theta_0 & \frac{-6E_l I_l}{m_0^2} \sin \theta_0 & \frac{2E_l I_l}{m_0} \\ -\frac{A_l E_l}{m_0} \sin^2 \theta_0 - \frac{12E_l I_l}{m_0^3} \cos^2 \theta_0 & \left(\frac{12E_l I_l}{m_0^3} - \frac{A_l E_l}{m_0}\right) \sin \theta_0 \cos \theta_0 & \frac{6E_l I_l}{m_0^2} \cos \theta_0 & \frac{A_l E_l}{m_0} \sin^2 \theta_0 + \frac{12E_l I_l}{m_0^3} \cos^2 \theta_0 & \left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & \frac{6E_l I_l}{m_0^2} \cos \theta_0 \\ \left(\frac{12E_l I_l}{m_0^3} - \frac{A_l E_l}{m_0}\right) \sin \theta_0 \cos \theta_0 & -\frac{A_l E_l}{m_0} \cos^2 \theta_0 - \frac{12E_l I_l}{m_0^3} \sin^2 \theta_0 & \frac{-6E_l I_l}{m_0^2} \sin \theta_0 & \left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & \frac{A_l E_l}{m_0} \cos^2 \theta_0 + \frac{12E_l I_l}{m_0^3} \sin^2 \theta_0 & \frac{-6E_l I_l}{m_0^2} \sin \theta_0 \\ -\frac{6E_l I_l}{m_0^2} \cos \theta_0 & \frac{6E_l I_l}{m_0^2} \sin \theta_0 & \frac{2E_l I_l}{m_0} & \frac{6E_l I_l}{m_0^2} \cos \theta_0 & \frac{-6E_l I_l}{m_0^2} \sin \theta_0 & \frac{4E_l I_l}{m_0} \end{bmatrix} \quad (A2)$$

The stiffness matrix of Element III is:

$$[k]_{III}^e = \begin{bmatrix} \frac{A_l E_l}{m_0} \sin^2 \theta_0 + \frac{12E_l I_l}{m_0^3} \cos^2 \theta_0 & -\left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & \frac{6E_l I_l}{m_0^2} \cos \theta_0 & -\frac{A_l E_l}{m_0} \sin^2 \theta_0 - \frac{12E_l I_l}{m_0^3} \cos^2 \theta_0 & \left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & \frac{6E_l I_l}{m_0^2} \cos \theta_0 \\ -\left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & \frac{A_l E_l}{m_0} \cos^2 \theta_0 + \frac{12E_l I_l}{m_0^3} \sin^2 \theta_0 & \frac{6E_l I_l}{m_0^2} \sin \theta_0 & \left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & -\frac{A_l E_l}{m_0} \cos^2 \theta_0 - \frac{12E_l I_l}{m_0^3} \sin^2 \theta_0 & \frac{6E_l I_l}{m_0^2} \sin \theta_0 \\ \frac{6E_l I_l}{m_0^2} \cos \theta_0 & \frac{6E_l I_l}{m_0^2} \sin \theta_0 & \frac{4E_l I_l}{m_0} & -\frac{6E_l I_l}{m_0^2} \cos \theta_0 & -\frac{6E_l I_l}{m_0^2} \sin \theta_0 & \frac{2E_l I_l}{m_0} \\ -\frac{A_l E_l}{m_0} \sin^2 \theta_0 - \frac{12E_l I_l}{m_0^3} \cos^2 \theta_0 & -\left(\frac{12E_l I_l}{m_0^3} - \frac{A_l E_l}{m_0}\right) \sin \theta_0 \cos \theta_0 & \frac{-6E_l I_l}{m_0^2} \cos \theta_0 & \frac{A_l E_l}{m_0} \sin^2 \theta_0 + \frac{12E_l I_l}{m_0^3} \cos^2 \theta_0 & -\left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & -\frac{6E_l I_l}{m_0^2} \cos \theta_0 \\ -\left(\frac{12E_l I_l}{m_0^3} - \frac{A_l E_l}{m_0}\right) \sin \theta_0 \cos \theta_0 & -\frac{A_l E_l}{m_0} \cos^2 \theta_0 - \frac{12E_l I_l}{m_0^3} \sin^2 \theta_0 & \frac{-6E_l I_l}{m_0^2} \sin \theta_0 & -\left(\frac{A_l E_l}{m_0} - \frac{12E_l I_l}{m_0^3}\right) \sin \theta_0 \cos \theta_0 & \frac{A_l E_l}{m_0} \cos^2 \theta_0 + \frac{12E_l I_l}{m_0^3} \sin^2 \theta_0 & \frac{-6E_l I_l}{m_0^2} \sin \theta_0 \\ \frac{6E_l I_l}{m_0^2} \cos \theta_0 & \frac{6E_l I_l}{m_0^2} \sin \theta_0 & \frac{2E_l I_l}{m_0} & -\frac{6E_l I_l}{m_0^2} \cos \theta_0 & \frac{-6E_l I_l}{m_0^2} \sin \theta_0 & \frac{4E_l I_l}{m_0} \end{bmatrix} \quad (A3)$$

Assembling of the global stiffness matrix as:

$$[K] = \begin{bmatrix} K_{11} & K_{12} & K_{13} & K_{14} & K_{15} & K_{16} & K_{17} & K_{18} & K_{19} \\ K_{21} & K_{22} & K_{23} & K_{24} & K_{25} & K_{26} & K_{27} & K_{28} & K_{29} \\ K_{31} & K_{32} & K_{33} & K_{34} & K_{35} & K_{36} & K_{37} & K_{38} & K_{39} \\ K_{41} & K_{42} & K_{43} & K_{44} & K_{45} & K_{46} & K_{47} & K_{48} & K_{49} \\ K_{51} & K_{52} & K_{53} & K_{54} & K_{55} & K_{56} & K_{57} & K_{58} & K_{59} \\ K_{61} & K_{62} & K_{63} & K_{64} & K_{65} & K_{66} & K_{67} & K_{68} & K_{69} \\ K_{71} & K_{72} & K_{73} & K_{74} & K_{75} & K_{76} & K_{77} & K_{78} & K_{79} \\ K_{81} & K_{82} & K_{83} & K_{84} & K_{85} & K_{86} & K_{87} & K_{88} & K_{89} \\ K_{91} & K_{92} & K_{93} & K_{94} & K_{95} & K_{96} & K_{97} & K_{98} & K_{99} \end{bmatrix}, \quad (A4)$$

Global thermal load is given as:

$$\{P\} = \begin{bmatrix} E_h A_h \alpha_h \Delta T + E_l A_l \alpha_l \Delta T \sin \theta_0 \\ -E_l A_l \alpha_l \Delta T \cos \theta_0 \\ 0 \\ -E_h A_h \alpha_h \Delta T - E_l A_l \alpha_l \Delta T \sin \theta_0 \\ -E_l A_l \alpha_l \Delta T \cos \theta_0 \\ 0 \\ 0 \\ 2EA\alpha\Delta T \cos \theta_0 \\ 0 \end{bmatrix} \quad (A5)$$

Applying boundary conditions leads to

$$\begin{bmatrix} K'_{11} & K'_{12} & K'_{13} \\ K'_{21} & K'_{22} & K'_{23} \\ K'_{31} & K'_{32} & K'_{33} \end{bmatrix} \begin{Bmatrix} u_1 \\ u_2 \\ u_3 \end{Bmatrix} = \begin{Bmatrix} P'_1 \\ P'_2 \\ P'_3 \end{Bmatrix}, \quad (A6)$$

where

$$K'_{11} = \frac{A_h E_h}{l_0} + \frac{A_l E_l}{m_0} \sin^2 \theta_0 + \frac{12 E_l I_l}{m_0^3} \cos^2 \theta_0, \tag{A7}$$

$$K'_{12} = K'_{21} = -\frac{A_h E_h}{l_0}, \tag{A8}$$

$$K'_{13} = K'_{31} = -\frac{A_l E_l}{m_0} \sin^2 \theta_0 - \frac{12 E_l I_l}{m_0^3} \cos^2 \theta_0 - \tan \varphi \left( \frac{12 E_l I_l}{m_0^3} - \frac{A_l E_l}{m_0} \right) \sin \theta_0 \cos \theta_0, \tag{A9}$$

$$K'_{22} = \frac{A_h E_h}{l_0} + \frac{A_l E_l}{m_0} \sin^2 \theta_0 + \frac{12 E_l I_l}{m_0^3} \cos^2 \theta_0, \tag{A10}$$

$$K'_{23} = K'_{32} = -\frac{A_l E_l}{m_0} \sin^2 \theta_0 - \frac{12 E_l I_l}{m_0^3} \cos^2 \theta_0 + \tan \varphi \left( \frac{12 E_l I_l}{m_0^3} - \frac{A_l E_l}{m_0} \right) \sin \theta_0 \cos \theta_0, \tag{A11}$$

$$K'_{33} = 2 \left( \frac{A_l E_l}{m_0} \sin^2 \theta_0 + \frac{12 E_l I_l}{m_0^3} \cos^2 \theta_0 \right) + 2 \tan^2 \varphi \left( \frac{A_l E_l}{m_0} \cos^2 \theta_0 + \frac{12 E_l I_l}{m_0^3} \sin^2 \theta_0 \right), \tag{A12}$$

$$\{P'\} = \begin{Bmatrix} E_h A_h \alpha_h \Delta T + E_l A_l \alpha_l \Delta T \sin \theta_0 \\ -E_h A_h \alpha_h \Delta T - E_l A_l \alpha_l \Delta T \sin \theta_0 \\ 2 E_l A_l \alpha_l \Delta T \cos \theta_0 \tan \varphi \end{Bmatrix}. \tag{A13}$$

**Appendix B**

The mechanical properties of the proposed metamaterials are investigated under tension using FEA. In the simulations, a  $4 \times 4$  unit cell assembled metamaterial is considered (see Fig. 9 (a)). A uniaxial displacement (10 mm) loading is applied on the top plate within the elastic limit, while the bottom plate is fixed along both the x and y directions. Three proposed metamaterial designs #A, #B, and #D are selected ( $A_h/A_l = 2.5$ ). Several material combinations including Al/Ti, Iron-Invar, Al6061/Ti-6Al-4V, PEGDA/Cu-PEGDA, Teflon/Acrylic and customized materials A/B are considered. The material properties of these constitutive materials are shown in Table 1. The FEA results are shown in Table 2.

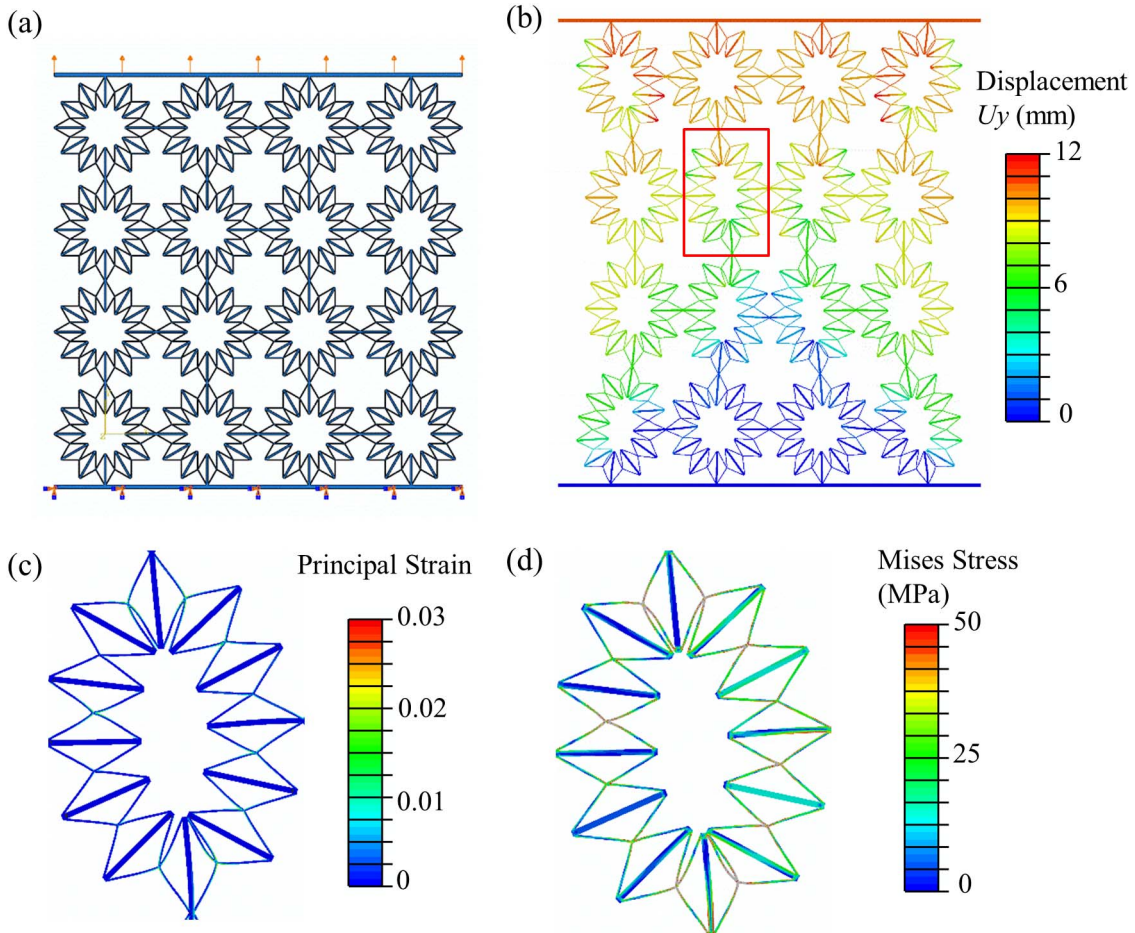


Fig. 9. Finite element model of the bi-material metamaterial #D ( $R_0 = 10$  mm,  $\theta_0 = 60^\circ$ ,  $n = 24$ ) under tension. (a) Undeformed configuration. (b) Deformed configuration. (c) FEA results of the maximum principal strain. (d) FEA results of the Mises stress.

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